

# Formation and Distribution of Phases during the Dehydration of Large Hydrargillite Floccules

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**Abstract**—The phase composition and phase distribution of large floccules of hydrargillite dehydration products obtained by heat treatment at temperatures from 250 to 500°C have been studied by thermal analysis, X-ray diffraction, and IR spectroscopy. Heat treatment in the range 250–300°C leads to the formation of macrocrystalline and microcrystalline boehmite particles. The macrocrystalline boehmite is formed from large hydrargillite crystals. The microcrystalline boehmite results from the “fragmentation” of the hydrargillite crystals. The IR spectra of the samples point to an increase in the number of hydroxyls with  $\nu(\text{OH}) = 3471 \text{ cm}^{-1}$  on the lateral faces of the hydrargillite crystals. The crystallite size of the hydrargillite crystals decreases by a factor of 1.8–2.3. In the floccules, a layer of the macrocrystalline boehmite surrounds a microcrystalline boehmite core. The outer surface is covered predominantly with  $\chi\text{-Al}_2\text{O}_3$ . Heat treatment at  $t \geq 350^\circ\text{C}$  leads to  $\gamma\text{-Al}_2\text{O}_3$  formation through sequential dehydration of microcrystalline and then macrocrystalline boehmite particles.

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## INTRODUCTION

The phase transformations of hydrargillite are well studied [1–9]. The paths of the phase transitions of hydrargillite have been determined at both low and high heating rates. Under near-equilibrium conditions [2] in air, the thermal decomposition of large particles proceeds through the formation of boehmite and  $\chi\text{-Al}_2\text{O}_3$ . Boehmite forms as a consequence of the hindered diffusion of the released water, resulting in hydrothermal conditions [2]. To ensure  $\chi\text{-Al}_2\text{O}_3$  crystallization, the water should be removed from the reaction zone, which is possible in particles  $<1 \text{ }\mu\text{m}$  in size or after large hydrargillite particles flake off, resulting in the formation of large pores [1, 2, 9]. At high heating rates in the course of thermochemical or thermocentrifugal activation [4, 5], hydrargillite converts to amorphous aluminum oxide.

At present, there is no general consensus as to the rate-limiting step in boehmite formation. According to Brown et al. [10], the rate of the process is limited by water diffusion and desorption, whereas in a number of reports [6–8, 11, 12] the formation and/or growth of boehmite crystal nuclei was thought to be rate controlling. The reported activation energies of the process range from 14 to 313 kJ/mol [7–9, 11–13], which can be accounted for in terms of different treatment and phase composition assessment conditions. Moreover, little is known about phase localization in large floccules of hydrargillite dehydration products [12]. Clearly, precise knowledge of the mechanisms underlying the formation of boehmite and alu-

minum oxides, the activation energies of the processes, and the phase distribution in the hydrargillite dehydration products in the form of large floccules is currently of special practical importance, because they find wide application as precursors or direct supports of catalysts for petrochemical processes, such as paraffin dehydrogenation and oxidative paraffin chlorination, in which hydrocarbons are brought into contact with a catalyst under fluidization conditions. Hydrargillite floccules range in size from 0.1 to 500  $\mu\text{m}$ , which allows one to synthesize catalysts of tailored granule size from such floccules. Hydrargillite is not used as a catalyst support because of its small specific surface area ( $<1 \text{ m}^2/\text{g}$ ).

One viable approach to the preparation of supports and catalysts is the dehydration of large hydrargillite floccules at atmospheric pressure. The properties of supports and catalysts can then be tuned and controlled even in the stage of hydrargillite phase transitions. As a consequence, it is of great current interest to study the effect of heat treatment conditions on the phase transformations in large floccules, the phase formation kinetics of boehmite and aluminum oxides, and their distribution in the floccules.

The purpose of this work was to study the formation and distribution of phases during the dehydration of large hydrargillite floccules at atmospheric pressure in air.

**Table 1.** Effect of the heat treatment conditions of hydrargillite on the crystallite size and composition of the boehmite phase

$t, ^\circ\text{C}$	$\tau, \text{min}$	Crystallite size, nm						wt %	
		HG		Bm + MBm mixture		calculated for MBm		Bm	MBm
		$D_{(002)}$	$D_{(110)}$	$D_{(020)}$	$D_{(120)}$	$D_{(020)}$	$D_{(120)}$		
—	—	67.5	55.0	—	—	—	—	—	—
250	60	65.8	57.0	50.2	38.6	—	—	100	—
	180	58.5	55.0	51.3	40.4	—	—	100	—
	300	57.6	55.0	50.2	42.3	—	—	100	—
	540	54.8	55.0	48.5	40.6	—	—	100	—
300	180	36.2	30.0	37.8	36.8	11.8	28.9	68.1	31.9
	240	34.0	29.9	36.7	34.9	10.9	23.9	66.4	33.9
	300	—	—	39.7	35.5	17.4	24.7	68.4	31.6
350	60	30.3	27.2	37.6	36.2	7.2	25.8	70.9	29.1
	180	—	—	38.0	33.0	5.1	12.8	73.3	25.7
	300	—	—	40.0	34.7	2.0	12.3	79.4	20.6
400	60	—	—	40.0	34.0	2.4	9.7	78.9	21.1
	180	—	—	41.6	40.4	—	—	100	—
	540	—	—	39.2	40.6	—	—	100	—
450	300	—	—	36.4	40.6	—	—	76.9	—
500	180	—	—	41.6	41.2	—	—	31.0	—

## EXPERIMENTAL

The starting material in this study was GD00 hydrargillite, ranging in floccule size from 40 to 180  $\mu\text{m}$ . It was dehydrated in a muffle furnace in the range 250 to 500 $^\circ\text{C}$  at a rate of 5 $^\circ\text{C}/\text{min}$  in air.

The phase composition of the products was determined by X-ray diffraction on a Shimadzu XRD-7000 diffractometer using long-wavelength  $\text{CuK}_\alpha$  radiation and a graphite monochromator (angular range  $2\theta = 5^\circ\text{--}90^\circ$ , scan step of 0.05 $^\circ$ ). The phases present were identified according to the presence of diffraction peaks of  $\gamma\text{-Al}(\text{OH})_3$  (ICSD no. 200 599),  $\gamma\text{-AlOOH}$  (ICSD no. 6162),  $\gamma\text{-Al}_2\text{O}_3$  (ICSD no. 66 559), and  $\chi\text{-Al}_2\text{O}_3$  (ICSD no. 13-373). The crystallite size was evaluated using the Scherrer formula. The error of crystallite size determination was  $\approx 10\%$ .

Thermal analysis (TA)<sup>1</sup> was carried out with a Netzsch STA-449C Jupiter simultaneous TG–DSC system in combination with an Aeolos QMS 403 quadrupole mass spectrometer in the range 30–1000 $^\circ\text{C}$  at a heating rate of 10 $^\circ\text{C}/\text{min}$  in flowing argon.

The particle size was determined using a Malvern Mastersizer 2000 laser diffraction analyzer, having a measurement range from 0.1 to 1000  $\mu\text{m}$ .

<sup>1</sup> The measurements were performed at the Federal Shared Facilities Center, Kazan Federal University, and were supported by the Federal Agency for Science and Innovations (A.V. Gerasimov).

IR spectra were taken on a Bruker VERTEX 70 FTIR spectrometer at room temperature, using samples pressed with KBr. The spectra were measured with a resolution of 1  $\text{cm}^{-1}$  and were averaged over 128 scans. Spectral lines were decomposed into Gaussian components and optimized using ORIGIN software.

Fine particles of hydrargillite dehydration products were obtained via the surface spalling of floccules in the course of attrition in a boiling bed in flowing air for 8 h.

## RESULTS AND DISCUSSION

The dehydration products contained two types of boehmite phase (Fig. 1): macrocrystalline (Bm) and microcrystalline (MBm). The oxide phase consisted of  $\chi\text{-Al}_2\text{O}_3$  or a mixture of  $\chi\text{-Al}_2\text{O}_3$  and  $\gamma\text{-Al}_2\text{O}_3$ . Hydrargillite (HG) converts to Bm at a temperature as low as 250 $^\circ\text{C}$  (Fig. 1). In contrast to the differential scanning calorimetry (DSC) curve of the parent hydrargillite, those of the samples show an increase in weight loss in the range 450–620 $^\circ\text{C}$ , due to Bm dehydration to  $\gamma\text{-Al}_2\text{O}_3$  (Fig. 2). The average crystallite sizes of Bm are  $\bar{D}_{(020)} = 50.5$  nm and  $\bar{D}_{(120)} = 40.5$  nm, which is close to the crystallite size of hydrargillite (Table 1). According to calculations of kinetic parameters [10], the rate-limiting step is the formation and/or growth of platelike nuclei of Bm crystals (Table 2), in agreement with previous results [6, 7, 11]. The kinetic curves for Bm formation can be described

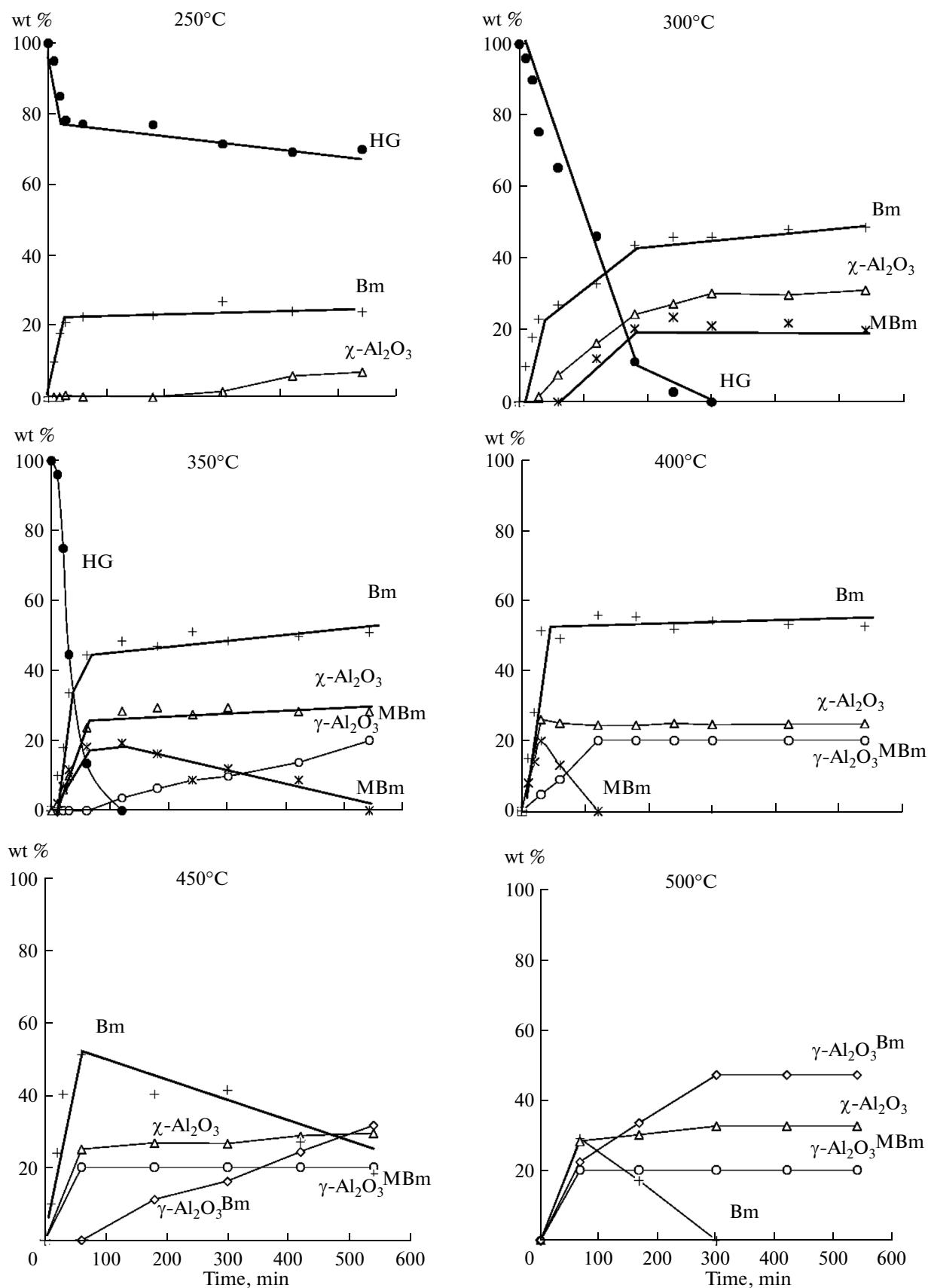


Fig. 1. Kinetic curves for hydrargillite conversion and formation of aluminum hydroxides and oxides.

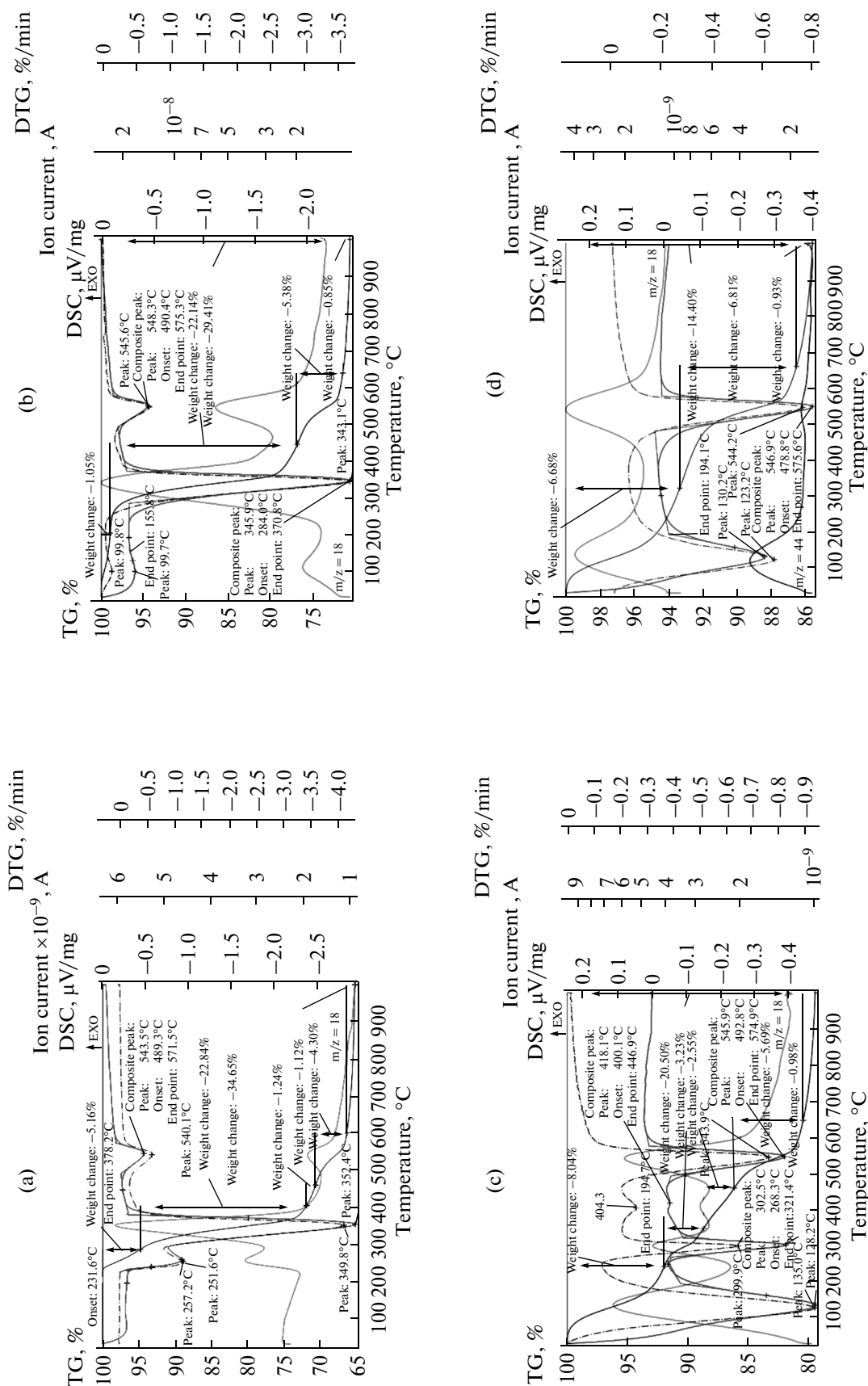


Fig. 2. Thermoanalytical curves of the parent hydargillite (a) and samples obtained by heat treatment at  $t = 250^\circ\text{C}$  for  $\tau = 420$  min (b), at  $t = 300^\circ\text{C}$  for  $\tau = 300$  min (c), and at  $t = 400^\circ\text{C}$  for  $\tau = 300$  min (d).

**Table 2.** Kinetic characteristics of various reactions during the thermal decomposition of hydrargillite (Avrami–Erofeev–Kolmogorov function [10])

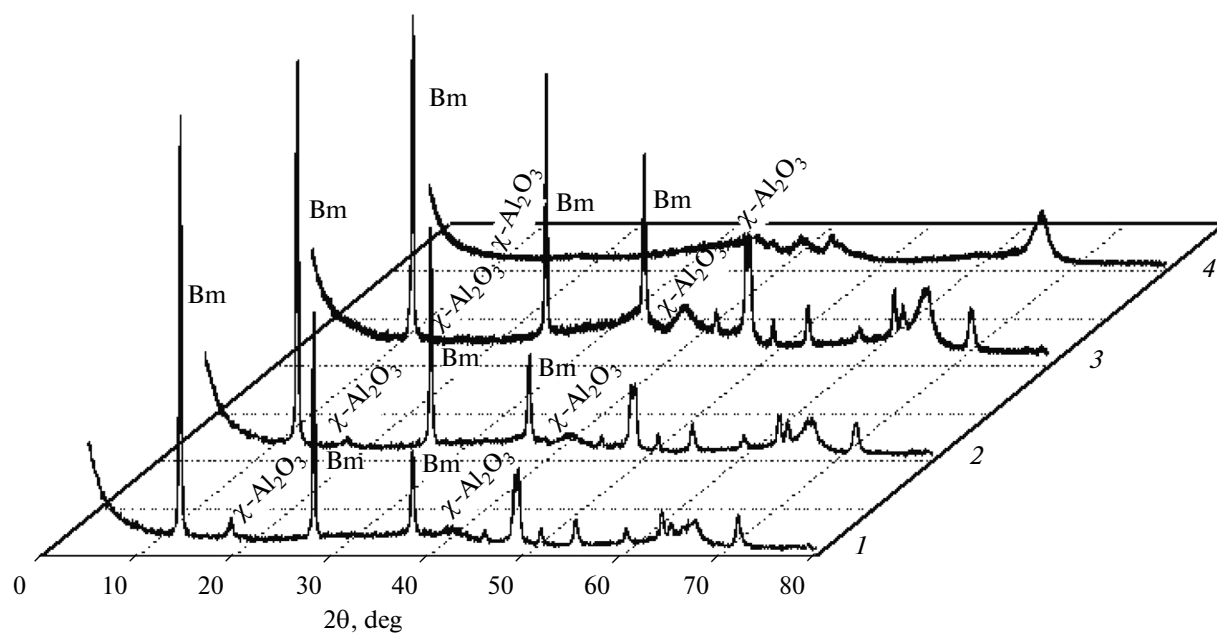
$t, \text{ }^{\circ}\text{C}$	$k, \text{ s}^{-1}$	$n$	$R^2$	$E_{\text{a}}, \text{ kJ/mol}$
HG dehydration				
200	$6.2 \times 10^{-6}$	1.5	0.8807	$74 \pm 4$
250	$2.2 \times 10^{-4}$	1.5	0.9903	
300	$2.4 \times 10^{-4}$	1.5	0.9938	
350	$4.1 \times 10^{-4}$	2.0	0.9698	
Bm formation				
200	$6.2 \times 10^{-6}$	1.5	0.8807	$62 \pm 4$
250	$1.4 \times 10^{-4}$	1.5	0.9695	
300	$2.2 \times 10^{-4}$	1.5	0.9387	
400	$5.8 \times 10^{-4}$	2.0	0.9478	
MBm formation				
300	$5.0 \times 10^{-5}$	1.5	0.9581	$94 \pm 7$
350	$1.4 \times 10^{-4}$	1.5	0.9961	
400	$3.7 \times 10^{-4}$	2.0	0.9996	
$\text{Al}_2\text{O}_3$ formation				
250	$8.5 \times 10^{-6}$	2.0	0.9430	$86 \pm 6$
350	$7.3 \times 10^{-5}$	2.0	0.9175	
400	$1.6 \times 10^{-4}$	1.5	0.9324	
500	$4.8 \times 10^{-4}$	2.0	0.9467	

by the Avrami–Erofeev–Kolmogorov equation, and the apparent activation energy ( $E_a$ ) is  $74 \pm 4$  kJ/mol (Table 2).

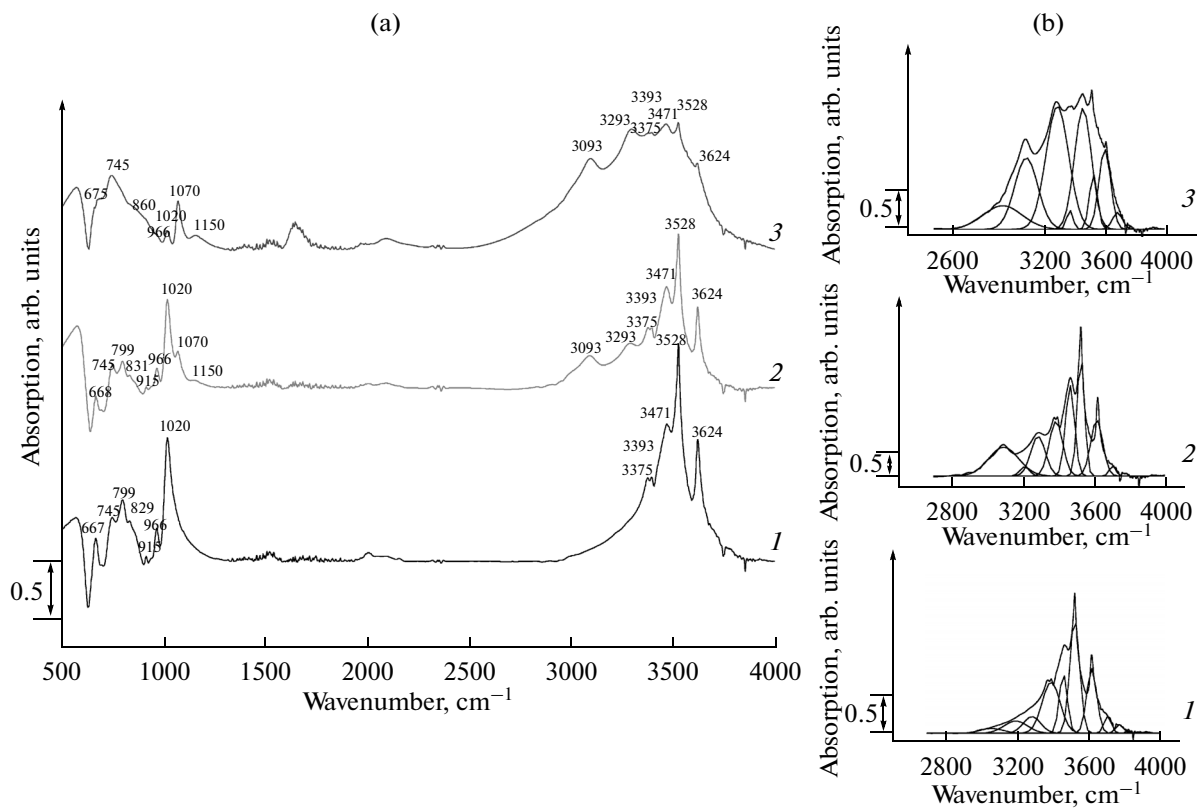
Quantitative MBm formation (Fig. 1) is evidenced by the presence of an additional, prominent endothermic peak in DSC curves in the range  $\approx 320$ – $460^\circ\text{C}$  (Fig. 2). Comparison of thermal analysis and X-ray diffraction data leads us to assign this peak to the dehydration of boehmite particles in  $\gamma$ - $\text{Al}_2\text{O}_3$ . In the X-ray diffraction patterns of the samples (Fig. 3), the reflections from boehmite broaden, but their position ( $2\theta$  angle) remains unchanged, suggesting that the content of structural water also does not vary (1.02 mol of  $\text{H}_2\text{O}$  per mole of  $\text{Al}_2\text{O}_3$ ). The crystallite size evaluated from the contribution of the forming aluminum hydroxide to this quantity is considerably smaller than that of Bm (Table 1), indicating that it has a microcrystalline nature (MBm). Calculations of kinetic parameters (Table 2) suggest a general mechanism for the formation and growth of nuclei of crystals of both boehmites, but  $E_a$  for boehmite formation exceeds that for Bm formation (Table 2), which is

caused by the preliminary “fragmentation” of the hydrargillite precursor crystals.

Fragmentation is also evidenced by IR spectroscopy data (Fig. 4) for the samples containing both MBm and residual HG. Fragmentation differs from delamination of hydrargillite particles [1]. In the former case, the crystallite size decreases considerably, whereas in the latter it remains almost unchanged. For example, in the IR spectra of the samples prepared at  $250^\circ\text{C}$  and containing  $\approx 70$  wt % unreacted hydrargillite, with only slight changes in crystallite size (Table 1), delamination causes a considerably larger decrease in the intensity of the absorption band at  $\nu(\text{OH}) = 3528 \text{ cm}^{-1}$ , arising from stretching vibrations of the interlayer hydrogen-bonded OH groups between the hydrargillite slabs [14], in comparison with the intensity of the absorption band at  $\nu(\text{OH}) = 3624 \text{ cm}^{-1}$ , due to vibrations of the terminal OH groups in the (001) plane of the crystals [14–16], and with that of the bands at  $\nu(\text{OH}) = 3471 \text{ cm}^{-1}$  and  $\nu(\text{OH}) = 3393$  and  $3375 \text{ cm}^{-1}$ , arising from vibrations



**Fig. 3.** X-ray diffraction patterns of the hydrargillite dehydration products obtained by heat treatment at (1)  $t = 300^\circ\text{C}$  for  $\tau = 300$  min, (2) at  $t = 400^\circ\text{C}$  for  $\tau = 300$  min, (3) at  $t = 450^\circ\text{C}$  for  $\tau = 300$  min, and (4) at  $t = 500^\circ\text{C}$  for  $\tau = 300$  min.



**Fig. 4.** (a) IR spectra of the parent hydrargillite (1) and thermal decomposition products obtained by heat treatment at  $t = 250^\circ\text{C}$  for  $\tau = 300$  min (2) and at  $t = 350^\circ\text{C}$  for  $\tau = 60$  min (3); (b) portions of the IR spectra after decomposition into Gaussians.

of the OH groups located on the lateral faces of the crystals and in the plane of the hydrargillite slabs, respectively [18]. At 300°C, as the fraction of converted hydrargillite approaches 100 wt % and that of the forming boehmite phase approaches its maximum level (Fig. 1), the reduction in the intensities of the absorption bands at  $\nu(\text{OH}) = 3624$  and  $3528 \text{ cm}^{-1}$  is accompanied by an increase in the intensity of the absorption bands at  $\nu(\text{OH}) = 3393$  and  $3375 \text{ cm}^{-1}$  and stabilization of the intensity of the band at  $\nu(\text{OH}) = 3471 \text{ cm}^{-1}$  (Fig. 5). The increase in the former case is due to the influence of the neighboring absorption bands arising from stretching vibrations of the OH groups of boehmite ( $\nu(\text{OH}) = 3093$  and  $3293 \text{ cm}^{-1}$ ), whose intensity rises by almost a factor of 1.5 (Fig. 4). In the latter case, there is no influence of boundary absorption bands: on the contrary, the intensity of the band at  $\nu(\text{OH}) = 3528 \text{ cm}^{-1}$  decreases. Analysis of the IR spectra after decomposition into Gaussians demonstrates that the area of the absorption band at  $\nu(\text{OH}) = 3471 \text{ cm}^{-1}$  increases (Fig. 6), suggesting an increase in the amount of OH groups on the lateral faces of the hydrargillite crystals because of their fragmentation. In addition, the crystallite size of the crystals decreases by a factor of 1.8–2.3 (Table 1), correlating with the area of the absorption band at  $\nu(\text{OH}) = 3471 \text{ cm}^{-1}$  (Fig. 7). Therefore, the hydrargillite crystal fragmentation products are MBm sources. They are concentrated primarily in the central part of the floccules, because the fine particles that flake off from the outer surface of the samples (Table 3) contain almost a factor of 2 less MBm than do the parent floccules.

In contrast to MBm, the  $\chi\text{-Al}_2\text{O}_3$  phase crystallizes predominantly on the outer surface of the floccules or on the surface of the forming pores [1, 11]. As distinct from the unprocessed floccules (Table 3), the fine particles in all of our samples are enriched in the  $\chi\text{-Al}_2\text{O}_3$  phase, whose content is higher by up to a factor of 4.5

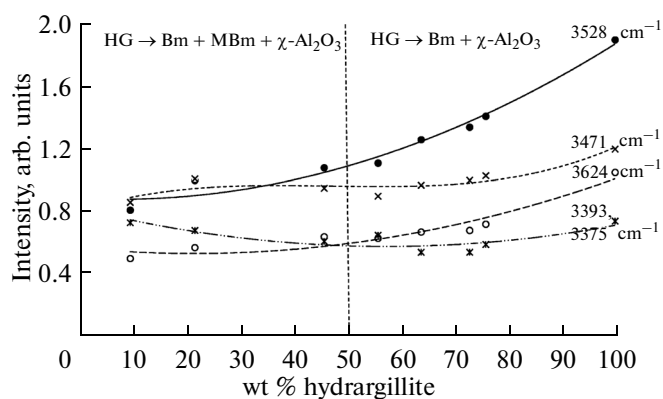


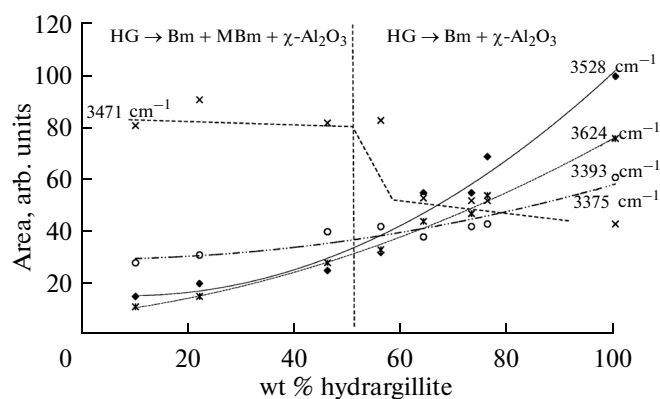
Fig. 5. Effect of hydrargillite concentration on the intensity of absorption bands in the IR spectra of the samples.

in the samples prepared at 250°C and by up to a factor of 1.4–1.6 in those prepared at 300–400°C. The  $\chi\text{-Al}_2\text{O}_3$  content of the dehydration products rises from 8 to 32 wt % in the range 250–300°C and then stabilizes between 350 and 500°C. The crystallite sizes of the crystals are  $D_{(440)} = 15.0 \text{ nm}$  and  $D_{(004)} = 32.1 \text{ nm}$ . The considerable induction period and slow rate of the process attest to a complex physicochemical nature of the formation and growth of platelike nuclei of the  $\chi\text{-Al}_2\text{O}_3$  phase, which requires a long time ( $\approx 60\text{--}300 \text{ min}$ ) to reach necessary changes in the structure of the precursor (Table 2, Fig. 1).

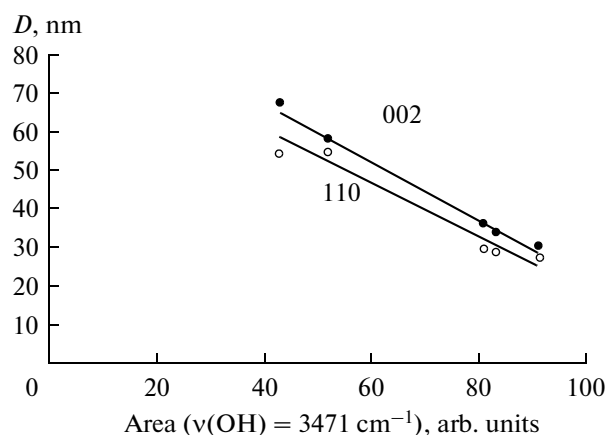
With increasing temperature, the percentage of oxide phases in the dehydration products increases on account of  $\text{Al}_2\text{O}_3^{\text{MBm}}$  and  $\gamma\text{-Al}_2\text{O}_3^{\text{Bm}}$ , resulting from the dehydration of MBm and Bm, respectively (Fig. 1). As in the case of the aluminum compounds considered above, the solid-state reaction is controlled by the formation and growth of platelike nuclei of  $\gamma\text{-Al}_2\text{O}_3$  (Table 2). The resulting  $\chi\text{-Al}_2\text{O}_3$  particles are

Table 3. Phase composition of floccules (I) and fine particles (II) obtained by attrition in a boiling bed

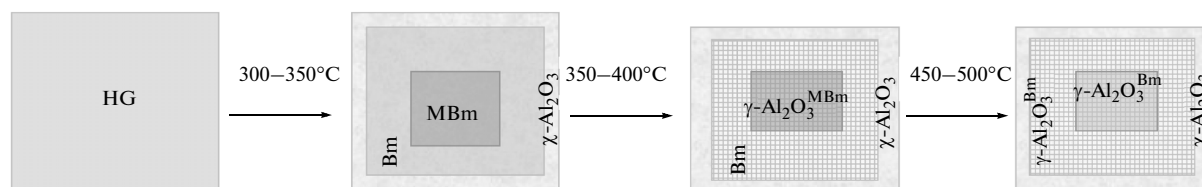
Phase	Composition (wt %) of samples produced by heat treatment							
	250°C, 540 min		300°C, 540 min		350°C, 240 min		400°C, 540 min	
	I	II	I	II	I	II	I	II
Hydrargillite	65.0	33.9	—	—	—	—	—	—
Bm	29.0	30.0	48.8	42.5	51.0	45.8	52.9	44.0
MBm	—	—	20.0	9.5	8.7	4.2	—	—
$\chi\text{-Al}_2\text{O}_3$	8.0	36.1	28.1	44.9	27.5	37.2	24.3	33.2
$\gamma\text{-Al}_2\text{O}_3$	—	—	3.1	3.1	12.8	12.8	22.8	22.8



**Fig. 6.** Effect of hydrargillite concentration on the areas of absorption bands in the IR spectra of the samples after decomposition into Gaussians.



**Fig. 7.** Correlation between the crystallite size of hydrargillite crystals and the area of the absorption band at  $\nu(\text{OH}) = 3471 \text{ cm}^{-1}$  in the IR spectra of the samples.



**Fig. 8.** Schematic of the phase distribution in flocules of hydrargillite decomposing at various dehydration temperatures in air.

much smaller than the  $\gamma\text{-Al}_2\text{O}_3$  particles (Table 1). The samples prepared at temperatures from 350 to 400°C have the form of multiphase flocules having an MBm and/or  $\gamma\text{-Al}_2\text{O}_3^{\text{MBm}}$  core, surrounded by a Bm layer and a  $\chi\text{-Al}_2\text{O}_3$  outer (peripheral) layer (Fig. 8). In the samples prepared in the range 450–500°C, the particles comprise a core consisting of  $\gamma\text{-Al}_2\text{O}_3^{\text{MBm}}$  crystals, a Bm and/or  $\gamma\text{-Al}_2\text{O}_3^{\text{Bm}}$  layer, and a  $\chi\text{-Al}_2\text{O}_3$  outer layer (Fig. 8). The outer shape of such multiphase particles remains pseudomorphic with the parent hydrargillite crystals, which in turn allows the floccule shape to persist.

## CONCLUSIONS

Heat treatment of hydrargillite flocules in air at temperatures from 250 to 500°C leads to the formation of the phases Bm, MBm,  $\chi\text{-Al}_2\text{O}_3$ ,  $\gamma\text{-Al}_2\text{O}_3^{\text{MBm}}$ , and  $\gamma\text{-Al}_2\text{O}_3^{\text{Bm}}$ , consisting of platelike crystals. The solid-state reactions are controlled by the formation and growth of platelike nuclei of the forming phases.

The fragmentation of hydrargillite crystals and the decrease in crystallite size lead to the formation of MBm at 300°C, which crystallizes mainly in the central part of the flocules. The MBm core is surrounded

by a Bm layer, which is covered with  $\chi\text{-Al}_2\text{O}_3$ , distributed predominantly on the outer surface of the flocules.

The flocules obtained at  $t \geq 350^\circ\text{C}$  have a core consisting of MBm and/or  $\gamma\text{-Al}_2\text{O}_3^{\text{MBm}}$  crystals and surrounded by a layer of platelike crystals of unreacted Bm and/or  $\gamma\text{-Al}_2\text{O}_3^{\text{Bm}}$ .

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